

(19)日本国特許庁 (JP)

(12) 公開特許公報 (A)

(11)特許出願公開番号

特開平8-171910

(43)公開日 平成8年(1996)7月2日

(51)Int.Cl. ⁶	識別記号	序内整理番号	F I	技術表示箇所
H 01 M 4/58				
C 01 G 53/00	A			
H 01 M 4/02	C			
10/40	Z			

審査請求 未請求 請求項の数2 O L (全5頁)

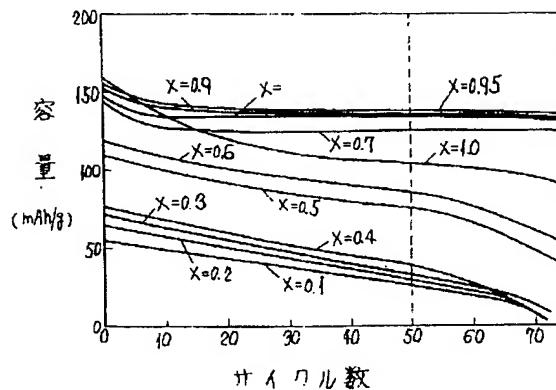
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(54)【発明の名称】 リチウム二次電池用正極活物質の製造法

(57)【要約】

【目的】 Niの一部をMnに確実に置換して一般式Li_{1-x}Mn_(1-x)O₂で表わされるリチウム含有複合酸化物の結晶構造をほぼ単一相とし、結晶完成度が高く結晶の崩壊がなく結晶内でLiが移動し易い安定した結晶場を得ることができる製造法を提供する。

【構成】 ニッケル塩とマンガン塩との混合溶液にアルカリ溶液を加えてニッケルとマンガンの水酸化物を共沈させることによって得たニッケルとマンガンの複合水酸化物を用いてリチウム含有複合酸化物を合成する。



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【特許請求の範囲】

【請求項1】リチウムとニッケルおよびマンガンを含むリチウム含有複合酸化物で、一般式 $L_iNi_xMn_{(1-x)}O_2$ で表わされる式中のx値を $0.95 \geq x \geq 0.70$ とする正極活物質の製造方法であり、マンガン塩とニッケル塩との混合水溶液にアルカリ溶液を加えてマンガンとニッケルの水酸化物を共沈させることによってマンガンとニッケルの複合水酸化物を得た後、水酸化リチウムなどのリチウム化合物と混合し、この混合物を焼成することを特徴とするリチウム二次電池用正極活物質の製造法。

【請求項2】マンガンとニッケルの複合水酸化物とリチウム化合物との混合物を、 600°C 以上 800°C 以下で焼成する請求項1記載のリチウム二次電池用正極活物質の製造法。

【発明の詳細な説明】

【0001】

【産業上の利用分野】本発明はリチウム二次電池の、とくにその正極活物質の製造法に関するものである。

【0002】

【従来の技術】近年、電子機器のポータブル化、コードレス化が急速に進んでおり、これらの駆動用電源として小形・軽量で、高エネルギー密度を有する二次電池への要望が高い。このような点で非水系二次電池、特にリチウム二次電池はとりわけ高電圧・高エネルギー密度を有する電池として期待が大きい。

【0003】このような中で L_iCoO_2 を正極に、炭素材料を負極に用いた電池が開発されている。 L_iCoO_2 の作動電位は L_i に対して4Vと高いため電池電圧が高くなるとともに、負極に炭素材料を用いてインタークレーション反応を利用しているため金属 L_i を負極に用いた場合の課題であったデンドライト状 L_i が負極上に析出することはなく電池の安全性を向上させることができる。

【0004】しかし、 Co の資源の問題とコストの問題から、 L_iCoO_2 に代わるリチウム含有複合酸化物の開発が進んでおり L_iNiO_2 などが注目されはじめた。 L_iNiO_2 ならびに L_iCoO_2 をはじめとするこの種のリチウム含有複合酸化物はいずれも高い電位を示し、かつインタークレーション反応の利用できる同じ六方晶系の結晶構造をもつ層状化合物であるため、正極活物質材料としてその期待が大きい。このような観点から、例えば L_iNiO_2 （米国特許第4302518号）、 $L_iNi_{2-y}O_2$ （特開平2-40861号公報）などの L_iNiO_2 に係るもの、あるいは $L_iNi_{1-x}Co_{1-x}O_2$ （特開昭63-299056号公報）や $L_iNi_{1-x}M_xO_2$ （但し、MはTi, V, Mn, Feのいずれか）などの L_iNiO_2 の Ni の一部を他の金属に置換したリチウム含有複合酸化物が提案されている。その他、 $A_xM_yNi_xO_2$ （但し、Aはアルカリ金属、Mは

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遷移金属、NはAl, In, Snの一種）（特開昭62-90863号公報）や $L_iM_xNi_{(1-x)}O_2$ （但し、MはFe, Co, Niの中から選ばれた少なくとも一種で、NはTi, V, Cr, Mnの中から選ばれた少なくとも一種）（特開平4-267053号公報）などのリチウム含有複合酸化物も提案されている。そしてこれらの活物質材料を用いて4V級の放電電位をもった高エネルギー密度のリチウム二次電池の開発が進められている。

【0005】

【発明が解決しようとする課題】これらのリチウム含有複合酸化物の中で L_iNiO_2 はリチウムに対し4Vの作動電位を示すので、正極活物質として用いると高エネルギー密度を有する二次電池が実現できる。しかし、電池の充放電サイクルの経過とともに電池容量が劣化し、50サイクル目では初期容量の65%まで低下し、良好な充放電サイクル特性が得られないという課題があった。

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【0006】このような課題に対し、上記に示すような Ni の一部を他の金属に置換したりチウム複合酸化物や多種の金属元素を同時に含むものなどが提案されてきた。しかし、 L_iNiO_2 の Ni の一部を他の金属に置換したものはサイクル可逆性が向上する一方、放電容量が小さくなり、かつ放電電圧も低くなる傾向にあり、本来要望されている高電圧、高エネルギー密度という特徴を減ずる結果となった。これらの中で Ni の一部をMnに置換したものはサイクル可逆性、放電容量、放電電圧のいずれも他のリチウム含有複合酸化物に比べると比較的良好であった。

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【0007】ここで、 L_iNiO_2 の Ni の一部をMnに置換した活物質の合成は、水酸化リチウムなどの L_i 化合物と水酸化ニッケルなどの Ni 化合物に二酸化マンガンや硝酸マンガンなどのMn化合物を加えて焼成する方法（以後、複合式合成法と呼ぶ）が一般的であった。この混合焼成法では Ni の一部をMnに確実に置換するためには少なくとも 800°C 以上の焼成温度が必要で、この温度以下ではX線回折を見る限り置換反応は完結しておらず、単一相を有する結晶完成度の高い化合物は得られなかった。

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【0008】しかし、 800°C 以上の高温で合成すると結晶中で L_i の入るべきサイトに Ni やMnが入り込んでしまい、結晶構造が乱れてしまいサイクル可逆性や放電容量が低下していた。このように L_iNiO_2 を基本にするリチウム含有複合酸化物を高温で焼成することはあまり好ましくなかった。

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【0009】本発明は、このような課題を解決するものであり、 Ni の一部をMnに確実に置換して一般式 $L_iNi_xMn_{(1-x)}O_2$ で表わされるリチウム含有複合酸化物の結晶構造をほぼ単一相とし、結晶完成度が高く結晶の崩壊がなく結晶内で L_i が移動し易い安定した結晶場を得ることができる製造法を提供するものである。

【0010】

【課題を解決するための手段】上記課題を解決するためには、本発明のリチウム二次電池用正極活物質の製造方法は、リチウムとニッケルおよびマンガン組成よりなる複合酸化物で、一般式 $L_i N_i \cdot Mn_{(1-x)} O_2$ で表わされ、式中の x 値を $0.95 \geq x \geq 0.70$ とする正極活物質の製造方法であり、マンガン塩とニッケル塩との混合水溶液にアルカリ溶液を加えてマンガンとニッケルの水酸化物を共沈させることによってマンガンとニッケルの複合水酸化物を得た後、水酸化リチウムなどのリチウム化合物と混合し、この混合物を $600^{\circ}C$ 以上 $800^{\circ}C$ 以下の温度範囲で焼成するものである。

【0011】

【作用】本発明の製造法では、マンガン塩とニッケル塩との混合溶液にアルカリ溶液を加えてマンガンとニッケルの水酸化物を共沈させることによりニッケルとマンガンの複合水酸化物（以下、 $Ni \cdot Mn$ 複合水酸化物）を得ているので、結晶構造が Ni の一部を Mn で確実に置換した固溶体レベルに至っており、X線回折でもほとんど単一相になつていて結晶完成度が極めて高いものとなつている。

【0012】そして、この $Ni \cdot Mn$ 複合水酸化物に L_i 塩を加えて焼成すると、結晶内で L_i が移動し易い結晶構造を有するリチウム含有複合酸化物を得ることができる。さらに本発明では焼成温度を $600^{\circ}C$ ~ $800^{\circ}C$ としているので結晶構造の乱れはない。

【0013】また、 Ni と Mn の混合原子価状態を形成して安定した結晶構造を得るために、少なくとも Ni の Mn への置換数は 0.05 以上必要である。しかし、 Ni の Mn への置換数が 0.30 を超えると結晶の歪みの増大や結晶構造の崩れの発生、および混合原子価状態の不釣り合いで L_i が動き難い状況を作り出して活物質の容量低下が著しくなる。

【0014】

【実施例】以下、本発明の実施例を図面を参照にしながら説明する。

【0015】まず、本発明の $Ni \cdot Mn$ 複合水酸化物の共沈による製造法を説明する。市販試薬の硫酸ニッケルを水に加え、飽和状態の硫酸ニッケル水溶液を作成し、これに所定量（目的の Mn/Ni 比に合わせて）の硫酸マンガンを加え、さらに水を加えて調整して硫酸ニッケルおよび硫酸マンガンを含む飽和水溶液を作成した。次いで、攪拌しながらこの水溶液に水酸化ナトリウムを溶解したアルカリ水溶液をゆっくりと加えていくと、 Ni と Mn の水酸化物の沈殿（共沈）が同時に始まった。十分にアルカリ溶液を加えて沈殿が終了したのを見極めた後、濾過して沈殿物を回収し水洗した。pHを測定しながら水洗を繰り返し、残存アルカリがほぼ無くなつたのを見極めた後、熱風空気 ($100^{\circ}C$) に設定した熱風乾燥器を用いた）で乾燥させた。

【0016】このようにして得られた $Ni \cdot Mn$ 複合水酸化物のX線回折パターンはきわめて単一相に近いものであり、元素分析の結果、ほぼ目的の比率で Mn と Ni を含んでいた。

【0017】なお、本実施例では共沈原材料の Ni 源として硫酸ニッケル、マンガン源として硫酸マンガンを用いたが、ニッケル源として硝酸ニッケル、マンガン源として硝酸マンガンなど、基本的には水溶液を作りうる塩であればいずれも使用可能である。また、アルカリ溶液としては水酸化ナトリウム水溶液を用いたが、水酸化カリウム水溶液、水酸化リチウム水溶液など他のアルカリ溶液であっても良い。

【0018】次いで、 L_i 化合物との焼成工程を説明する。 L_i 化合物としては水酸化リチウムを用い、上記共沈で得られた $Ni \cdot Mn$ 複合水酸化物に Mn と Ni の原子数の和と L_i の原子数が等量になるよう加えてボールミルで粉碎しながら十分混合し、この複合物をアルミニナ製るつぼに入れ酸素中において $550^{\circ}C$ で 20 時間で 1段目の焼成をした後、 $750^{\circ}C$ で 2 時間で 2段目の焼成をした。焼成後室温までゆっくりと冷却し、粉碎したものを正極活物質粉末とした。

【0019】 Mn/Ni 比の異なるいくつかの $Mn \cdot Ni$ 複合水酸化物について合成を試みた結果、活物質の組成を示す一般式 $L_i N_i \cdot Mn_{(1-x)} O_2$ の x 値が 0.7 以上であるとこのリチウム含有複合酸化物のX線回折パターンが単一相で得られた。しかし、 x 値が 0.7 未満になるとX線パターンはほぼ単一相ではあるものの、ピーク強度が弱まり結晶性が低下する傾向があった。さらに、 x 値が 0.5 を下回ると、六方晶系の層状構造が崩れていた。

【0020】そして、この正極活物質 100 重量部に対してアセチレンブラックを 5 重量部加え十分に混合した後、この混合物を N-メチルピロリジン (NMP) の溶媒に接着剤のポリフッ化ビニリデン (PVDF) を溶解した液で練りペーストとした。なお、PVDF の量は正極活物質 100 重量部に対して 4 重量部となるように調整した。次いで、このペーストをアルミ箔の片面に塗着した後、乾燥して圧延し極板とした。図 1 は本発明の実施例に用いたコイン形リチウム二次電池の縦断面図である。図 1 において、正極 1 は前記極板を円板状に打ち抜いたもので、正極ケース 2 の内側に設置したものである。また、負極 3 は金属リチウムをステンレス鋼製ネット 5 上に圧着したもので、封口板 4 の内側にスポット溶接されている。正極 1 と負極 3 の間にはポリプロピレン製セバレータ 6 が配されており電解液 7 が注液されている。また、ポリプロピレン製ガスケット 8 を介して密封した。なお、電解液には 1 モルの六フッ化リン酸リチウム ($L_i PF_6$) を炭酸エチレン (EC) と炭酸ジエチル (DEC) の混合溶媒中に溶かしたもの用いた。

【0021】そして、一般式 $L_i N_i \cdot Mn_{(1-x)} O_2$ で

表わされる正極活物質の x の値を 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.7, 0.8, 0.9, 0.95, 1.0 とし、これらを用いて上記と同様の方法でコイン形電池を作製した。なお、 $x = 1.0$ は Mn を含まない LiNiO_2 である。ついで、これらの電池を用いて充放電サイクル寿命試験を行った。充放電条件は、室温 (20°C) で正極に対して 0.5 mA/c m^2 の定電流で充放電し、充電終止電圧を 4.3 V、放電終止電圧を 3.0 V として行った。

【0022】図2は充放電サイクル試験の結果で、 $x = 0.1 \sim 0.4$ の範囲の正極活物質は初期容量が 50 ~ 80 mAh/g と小さく、かつサイクル劣化も大きく、50サイクル目で初期容量の 50%まで低下し、その後も劣化が進んだ。

【0023】 $x = 0.5 \sim 0.6$ の範囲の正極活物質は初期容量が 110 ~ 120 mAh/g であったが、サイクル劣化が大きく、50サイクル目で初期容量の 70%まで低下し、その後も劣化が進んだ。

【0024】 $x = 0.7 \sim 0.95$ の範囲の正極活物質は初期容量が 140 ~ 150 mAh/g と大きく、かつサイクル劣化も小さく、50サイクル目で初期容量の 90%を維持しているとともにそれ以後のサイクルの繰り返しにおいても容量低下がほとんど見られなかった。ところが、 $x = 1.0$ で Mn を含まない正極活物質は初期容量こそ 150 mAh/g 以上のものが得られるもののサイクル劣化は大きく、50サイクル目で初期容量の 65%まで低下し、その後も劣化が進んだ。

【0025】以上の結果からも明らかなように、共沈により調整した $\text{Ni} \cdot \text{Mn}$ 複合水酸化物を用いて合成した活物質 $\text{LiNi}_{x} \text{Mn}_{(1-x)} \text{O}_2$ における x の値は 0.7 ~ 0.95 の範囲のものが好ましい。

【0026】次に、一般式 $\text{LiNi}_{x} \text{Mn}_{(1-x)} \text{O}_2$ で表わされる正極活物質の x の値が 0.70 ~ 0.95 の範囲のものについて焼成温度を変える検討を行った。1段目の焼成である 550°C 20 時間の工程は上記と同様に行い、その後の焼成について焼成温度を 550°C, 600°C, 650°C, 700°C, 750°C, 800°C, 850°C, 900°C とした。そして、これらの正極活物質を用いて上記と同様の電池を構成し、上記と同様の条件の充放電サイクル試験を行った。図3にこの結果を示す。なお、上記式中の x 値は 0.8 とした。

【0027】図3からも明らかなように、焼成温度を 600°C ~ 800°C として合成した活物質が初期容量、ならびにサイクル特性も良好で、550°C のものは初期容量、サイクル性とともに不十分で、850°C ~ 900°C のものは初期容量が若干小さくなり、サイクル劣化も大きくなつた。

【0028】以上の結果より、正極活物質の焼成温度は 600°C ~ 800°C が良いが、800°Cになるとサイクル劣化が若干大きくなり、初期容量も若干小さめにな

り、600°Cになると初期容量は良好なもののサイクル劣化が若干大きくなるので、650°C ~ 750°C が好ましい。

【0029】本実施例では、 x 値が 0.8 の場合のものについて述べたが、 x 値が 0.70 ~ 0.95 の範囲のものについてそれぞれ同様の焼成温度に関する検討を行った結果、 $x = 0.8$ の場合と同様の傾向を示す結果が得られた。

【0030】従来の混合式合成法を用いて、 $\text{LiNi}_{0.8} \text{Mn}_{0.2} \text{O}_2$ の組成を有する正極活物質を合成した。まず、水酸化ニッケルと水酸化リチウムと水酸化マンガンとを $\text{Ni} : \text{Mn} : \text{Li}$ の原子比が 0.8 : 0.2 : 1.0 となるように秤量し、ボールミルで粉碎しながら混合し、混合物をアルミナるつばに入れ酸素中において 550°C, 20 時間で 1段目の焼成をした後、750°C, 2 時間で 2段目の焼成をした。焼成後室温までゆっくりと冷却し、粉碎したものを正極活物質とした。この活物質の X 線回折パターンは単一相にならず、複数相が存在するものとなった。そこで、2段目の焼成温度を 650°C ~ 900°C の範囲で変える検討を行った結果、焼成温度を 800°C 以上にすることによって単一相が得られるようになった。

【0031】図4は上記各温度によって作成した活物質を用いて上記と同様の充放電サイクル試験を行った結果であるが、焼成温度を 650°C ~ 750°C とした正極活物質は単一相の結晶構造が得られなく、初期容量も小さいとともにサイクル劣化が著しく、ほとんど 50 サイクル時点で初期容量の 50%以下に低下し、その後も劣化が進んだ。

【0032】一方、800°C 以上の焼成温度で合成した活物質は結晶構造が単一相となり、50 サイクル時点での初期容量の 80%を維持するが、容量値が 100 mAh/g 以下になりさらに減少した。

【0033】

【発明の効果】以上のように本発明のリチウム二次電池用正極活物質の製造法では、ニッケル塩とマンガン塩との混合溶液にアルカリ溶液を加えてニッケルとマンガンの水酸化物を共沈させることによりニッケルとマンガンの複合水酸化物（以下、 $\text{Ni} \cdot \text{Mn}$ 複合酸化物）を得ているので、結晶構造が Ni の一部を Mn で確実に置換した固溶体レベルに至っており、X線回折でもほとんど単一相になっていて結晶完成度が極めて高いものとなっている。そして、この $\text{Ni} \cdot \text{Mn}$ 複合水酸化物に Li 塩を加えて焼成すると、結晶内で Li が移動し易い結晶構造を有するリチウム含有複合酸化物を得ることができ、容量が大きくサイクル特性に優れた正極活物質を得ることができる。

【図面の簡単な説明】

【図1】本実施例に用いたコイン形リチウム二次電池の断面図

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【図2】 x 値を変化させたときの正極活物質の容量とサイクル数との関係を示す図

【図3】焼成温度を変化させたときの正極活物質の容量とサイクル数との関係を示す図

【図4】従来の製造法により合成した正極活物質の容量とサイクル数との関係を示す図

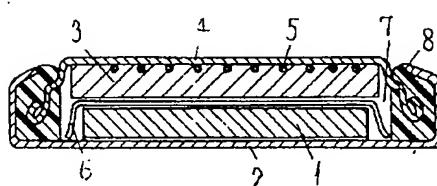
【符号の説明】

1 正極

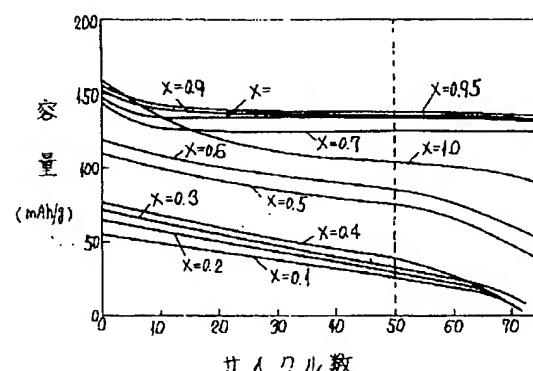
- 2 正極ケース
- 3 負極
- 4 封口板
- 5 ステンレス鋼製ネット
- 6 セパレータ
- 7 電解液
- 8 ガスケット

8

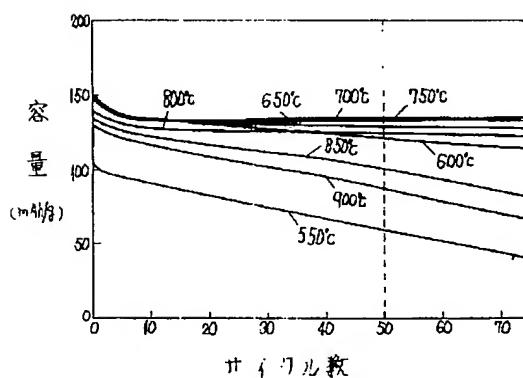
【図1】



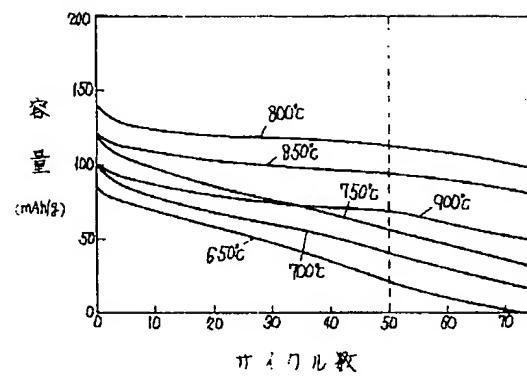
【図2】



【図3】



【図4】



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PATENT ABSTRACTS OF JAPAN

(11) Publication number : 08-171910

(43) Date of publication of application : 02.07.1996

(51)Int.CI. H01M 4/58
C01G 53/00
H01M 4/02
H01M 10/40

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(54) MANUFACTURE OF POSITIVE ELECTRODE ACTIVE MATERIAL FOR LITHIUM SECONDARY BATTERY

(57) Abstract:

PURPOSE: To make a crystal structure to be of a single phase, and thereby obtain an oxide high in crystal finishing degree by mixing coprecipitated hydrooxide which is obtained by adding alkali solution to the mixed water solution of manganese salts and nickel salts, with lithium compounds, and thereby sintering a products thus obtained at a specified temperature.

CONSTITUTION: In the manufacture of a composite oxide for positive electrode active material indicated by a formula, alkali water solution is added first to the mixed water solution of manganese salt and nickel salt the amounts of which are divided by a specified volumetric ratio, so as to be coprecipitated, the product is then filtered and cleaned with water, and a manganese nickel hydrooxide is thereby obtained thereafter. It is proved by a X ray diffraction that the product is roughly of a single phase, and thereby its

crystal finishing degree is high. Next, a specified amount of a lithium compound such as lithium hydroooxide and the like is mixed to the aforesaid composite hydroooxide so as to be sintered, and a composite oxide containing lithium with a crystal structure in which Li

is easily moved, is thereby obtained. Furthermore, since its sintering temperature is set to be 600 to 800°C, there occurs no disturbance in a crystal structure. By this constitution, the positive electrode active material can be obtained, which is high in capacity and excellent in cyclic characteristics. (in the formula, x is defined by $0.95 \geq x \geq 0.70$).

LEGAL STATUS

[Date of request for examination] 21.11.1997

[Date of sending the examiner's decision of rejection]

[Kind of final disposal of application other than the examiner's decision of rejection or application converted registration]

[Date of final disposal for application]

[Patent number] 3008793

[Date of registration] 03.12.1999

[Number of appeal against examiner's decision of rejection]

[Date of requesting appeal against examiner's decision of rejection]

[Date of extinction of right]

CLAIMS

[Claim(s)]

[Claim 1] With the lithium content multiple oxide containing a lithium, nickel, and manganese General formula LiNixMn ($1-x$) It is the manufacture approach of the positive active material which sets to $0.95 >= x >= 0.70$ x values in the formula expressed with O₂. After obtaining the compound hydroxide of manganese and nickel by adding an alkali solution to the mixed water solution of manganese salt and nickel salt, and carrying out coprecipitation of the hydroxide of manganese and nickel, The manufacturing method of the positive active material for lithium secondary batteries characterized by mixing with lithium compounds, such as a lithium hydroxide, and calcinating this mixture.

[Claim 2] The manufacturing method of the positive active material for lithium secondary batteries according to claim 1 which calcinates the mixture of manganese, the compound hydroxide of nickel, and a lithium compound below 600 degrees C or more 800 degrees C.

DETAILED DESCRIPTION

[Detailed Description of the Invention]

[0001]

[Industrial Application] This invention is the thing especially about the manufacturing method of the positive active material of a lithium secondary battery.

[0002]

[Description of the Prior Art] In recent years, portable-izing of electronic equipment and cordless-ization are progressing quickly, it is small and lightweight as these power sources for a drive, and the request to the rechargeable battery which has a high energy consistency is high. A nonaqueous rechargeable battery, especially a lithium secondary battery have great expectation at such a point as a cell which especially has a high voltage and a high energy consistency.

[0003] The cell which thus, used LiCoO₂ for the positive electrode and used the carbon material for the negative electrode is developed. 4V, since it is high, while cell voltage becomes high to Li, since the action potential of LiCoO₂ used carbon material for the negative electrode and uses the intercalation reaction, the shape of a dendrite Li which was a technical problem at the time of using Metal Li for a negative electrode cannot deposit on a negative electrode, and it can raise the safety of a cell.

[0004] However, development of the lithium content multiple oxide which replaces LiCoO₂ is progressing from the problem of the resource of Co, and the problem of cost, and LiNiO₂ etc. began to attract attention. Since these kinds including LiNiO₂ and LiCoO₂ of lithium content multiple oxide [each] is a stratified compound with the crystal structure of the same hexagonal system which shows high potential and can use an intercalation reaction, that expectation is great as a positive-active-material ingredient.

The lithium content multiple oxide which permuted some nickel of LiNiO_{(s)2}, such as a thing which starts LiNiO_{(s)2}, such as LixNiO₂ (U.S. Pat. No. 4302518) and LiyNi_{2-y}O₂

(JP,2-40861,A), for example, or $\text{Li}_{\text{x}}\text{Ni}_{\text{x}}\text{Co}_{1-\text{x}}\text{O}_2$ (JP,63-299056,A), $\text{Li}_{\text{x}}\text{Ni}_{1-\text{x}}\text{M}_{\text{x}}\text{O}_2$ (however, M Ti, V, Mn, or Fe), by other metals from such a viewpoint is proposed. In addition, lithium content multiple oxides, such as $\text{A}_{\text{x}}\text{M}_{\text{y}}\text{N}_{\text{z}}\text{O}_2$ (however, A alkali metal and M transition metals and N a kind of aluminum, In, and Sn) (JP,62-90863,A) and $\text{L}_{\text{x}}\text{M}_{\text{y}}\text{N}_{\text{z}}\text{O}_2$ (however, M was chosen from Fe, Co, and nickel it is a kind at least and N was chosen from Ti, V, Cr, and Mn at least a kind) (JP,4-267053,A), are also proposed. And development of the lithium secondary battery of the high energy consistency which had 4V class discharge potential using these active material ingredients is furthered.

[0005]

[Problem(s) to be Solved by the Invention] Since LiNiO_2 shows the action potential of 4V to a lithium in these lithium content multiple oxides, if it uses as positive active material, the rechargeable battery which has a high energy consistency is realizable. However, cell capacity deteriorated with progress of the charge-and-discharge cycle of a cell, by 50 cycle eye, it fell to 65% of initial capacity, and the technical problem that a good charge-and-discharge cycle property was not acquired occurred.

[0006] What contains in coincidence the metallic element of the lithium multiple oxide and variety which permuted some nickel as shown above by other metals has been proposed to such a technical problem. However, while cycle reversibility of what permuted some nickel of LiNiO_2 by other metals improved, it is in the inclination for discharge capacity to become small and for discharge voltage to also become low, and resulted in reducing the description of the high voltage and high energy consistency which are originally demanded. Both cycle reversibility discharge capacity and the discharge voltage of what permuted some nickel by Mn in these were comparatively good compared with other lithium content multiple oxides.

[0007] Here, composition of the active material which permuted some nickel of LiNiO_2 by Mn had the common method (it is henceforth called a compound expression synthesis method) of adding and calcinating Mn compounds, such as a manganese dioxide and manganese nitrate, to Li compounds, such as a lithium hydroxide, and nickel compounds, such as nickel hydroxide. By this mixed calcinating method, in order to permute some nickel by Mn certainly, the burning temperature of at least 800 degrees C or more was required, and as long as the X diffraction was seen, the substitution reaction was not completed, and the compound with the high crystal completeness which has a single phase was not obtained below at this temperature.

[0008] However, when compounded at the elevated temperature 800 degrees C or more, nickel and Mn entered the site into which Li should go in a crystal, the crystal structure was confused, and cycle reversibility and discharge capacity were falling. Thus, it was not so desirable to have calcinated the lithium content multiple oxide based on LiNiO_2 at an elevated temperature.

[0009] This invention offers the manufacturing method which can obtain the stable crystal field where such a technical problem is solved, the crystal structure of a lithium content multiple oxide which permutes some nickel by Mn certainly and is expressed with general formula $\text{Li}_{\text{x}}\text{Ni}_{\text{x}}\text{Mn}(1-\text{x})\text{O}_2$ is mostly made into a single phase, there is no collapse of a crystal highly [crystal completeness], and Li tends to move within a crystal.

[0010]

[Means for Solving the Problem] In order to solve the above-mentioned technical

problem, the manufacture approach of the positive active material for lithium secondary batteries of this invention It is expressed by general formula $\text{LiNixMn}(1-x)\text{O}_2$ by the multiple oxide which consists of a lithium, nickel, and a manganese presentation. It is the manufacture approach of the positive active material which sets x values in a formula to $0.95 >= x >= 0.70$. After obtaining the compound hydroxide of manganese and nickel by adding an alkali solution to the mixed water solution of manganese salt and nickel salt, and carrying out coprecipitation of the hydroxide of manganese and nickel, It mixes with lithium compounds, such as a lithium hydroxide, and this mixture is calcinated in 600-degree-C or more temperature requirement 800 degrees C or less.

[0011]

[Function] According to the manufacturing method of this invention, since the compound hydroxide (following and nickel-Mn compound hydroxide) of nickel and manganese has been obtained by adding an alkali solution to the mixed solution of manganese salt and nickel salt, and carrying out coprecipitation of the hydroxide of manganese and nickel, it has resulted in the solid-solution level which permuted some nickel certainly from Mn, and also by the X diffraction, the crystal structure almost has a single phase and has become what has very high crystal completeness.

[0012] And if Li salt is added and calcinated to this nickel-Mn compound hydroxide, the lithium content multiple oxide which has the crystal structure which Li tends to move within a crystal can be obtained. Since burning temperature is furthermore made into 600 degrees C - 800 degrees C by this invention, there is no turbulence of the crystal structure.

[0013] Moreover, in order to acquire the crystal structure which formed the mixed valence condition of nickel and Mn and was stabilized, the number of permutations to Mn of nickel at least is [0.05 or more] required. However, if the number of permutations to Mn of nickel exceeds 0.30, increase of distortion of a crystal, generating of collapse of the crystal structure, and a mixed valence condition are disproportionate, the situation of being hard to move Li will be made and capacity deterioration of an active material will become remarkable.

[0014]

[Example] Hereafter, the example of this invention is explained for a drawing, making it reference.

[0015] First, the manufacturing method by the coprecipitation of the nickel-Mn compound hydroxide of this invention is explained. The nickel sulfate of a commercial reagent was added to water, the nickel-sulfate water solution of a saturation state was created, the manganese sulfate of the specified quantity (doubling with the target Mn/nickel ratio) was added to this, and the saturated water solution which adds and adjusts water further and contains a nickel sulfate and a manganese sulfate was created. Subsequently, when the alkali water solution which dissolved the sodium hydroxide in this water solution was added slowly, stirring, precipitate (coprecipitation) of the hydroxide of nickel and Mn started in coincidence. It filtered, and precipitate was collected and rinsed, after having discerned that fully added the alkali solution and precipitate was completed. Rinsing is repeated measuring pH, and after discerning that residual alkali was lost mostly, it was made to dry with hot blast air (for the air forced oven set as 100 degrees C to have been used).

[0016] Thus, the X diffraction pattern of the obtained nickel-Mn compound hydroxide

was very close to the single phase, and contained Mn and nickel by the target ratio mostly as a result of elemental analysis.

[0017] In addition, although the manganese sulfate was used as a nickel sulfate and a source of manganese as a source of nickel of a coprecipitation raw material in this example, all are usable if manganese nitrate etc. is the salt which can make a water solution fundamentally as nickel nitrate and a source of manganese as a source of nickel. Moreover, although the sodium-hydroxide water solution was used as an alkali solution, you may be other alkali solutions, such as a potassium-hydroxide water solution and a lithium-hydroxide water solution.

[0018] Subsequently, a baking process with Li compound is explained. The 2nd step of baking was carried out at 750 degrees C in 2 hours after [while the ball mill ground in addition so that the sum of the atomic number of Mn and nickel and the atomic number of Li might become the nickel-Mn compound hydroxide obtained by the above-mentioned coprecipitation ana, using a lithium hydroxide as a Li compound,] mixing enough, putting this composite into the crucible made from an alumina and carrying out the 1st step of baking at 550 degrees C into oxygen in 20 hours. It cooled slowly to the after [baking] room temperature, and what was ground was used as positive-active-material powder.

[0019] As a result of trying composition about some Mn-nickel compound hydroxides with which Mn/nickel ratios differ, the X diffraction pattern of this lithium content multiple oxide was obtained with the single phase as x values of general formula $\text{LiNixMn}(1-x)\text{O}_2$ which show the presentation of an active material are 0.7 or more. However, although the X-ray pattern was a single phase mostly when x values became less than 0.7, there was an inclination for peak intensity to become weaker and for crystallinity to fall. Furthermore, when x values were less than 0.5, the layer structure of hexagonal system had collapsed.

[0020] And after mixing acetylene black to this positive-active-material 100 weight section in 5 weight ***** 10 minutes, this mixture was scoured with the liquid which dissolved the polyvinylidene fluoride (PVDF) of a binder in the solvent of N-methyl pyrrolidinone (NMP), and it considered as the paste. In addition, the amount of PVDF was adjusted so that it might become 4 weight sections to the positive-active-material 100 weight section. Subsequently, after plastering one side of aluminum foil with this paste, it dried and rolled out and considered as the plate. Drawing 1 is drawing of longitudinal section of the coin form lithium secondary battery used for the example of this invention. In drawing 1, a positive electrode 1 is what pierced said plate to disc-like, and is installed inside the positive-electrode case 2. Moreover, a negative electrode 3 is what stuck the metal lithium by pressure on the network 5 made from stainless steel, and spot welding is carried out inside the obturation plate 4. The separator 6 made from polypropylene is allotted between the positive electrode 1 and the negative electrode 3, and the electrolytic solution 7 is poured in. Moreover, it sealed through the gasket 8 made from polypropylene. In addition, what melted the one-mol 6 phosphorus-fluoride acid lithium (LiPF_6) in ethylene carbonate (EC) and the mixed solvent of diethyl carbonate (DEC) was used for the electrolytic solution.

[0021] and the value of x of positive active material expressed with general formula $\text{LiNixMn}(1-x)\text{O}_2$ -- 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.7, 0.8, 0.9, and 0. -- it was referred to as 95 and 1.0 and the coin form cell was produced by the same approach as the above using

these. In addition, $x= 1.0$ is LiNiO_2 which does not contain Mn. Subsequently, charge-and-discharge cycle life test was performed using these cells. The charge and discharge of the charge-and-discharge conditions were carried out according to the constant current of 0.5 mA/cm² to the positive electrode at the room temperature (20 degrees C), and they performed 4.3V and discharge final voltage for the charge termination electrical potential difference as 3.0V.

[0022] The positive active material of the range of $x=0.1-0.4$ had an initial capacity as small as 50 - 80 mAh/g, and cycle degradation of drawing 2 was also large at the result of a charge-and-discharge cycle trial, it fell to 50% of initial capacity by 50 cycle eye, and, also after that, degradation progressed.

[0023] The positive active material of the range of $x=0.5-0.6$ had large cycle degradation, although initial capacity was 110 - 120 mAh/g, and it fell to 70% of initial capacity by 50 cycle eye, and, also after that, degradation progressed.

[0024] While the positive active material of the range of $x=0.7-0.95$ had an initial capacity as large as 140 - 150 mAh/g, and cycle degradation was also small and maintaining 90% of initial capacity by 50 cycle eye, also in the repeat of the cycle after it, a capacity fall was hardly seen. However, although, as for the positive active material which does not contain Mn by $x= 1.0$, the thing of 150 or more mAh/g was obtained just for initial capacity, cycle degradation was large, it fell to 65% of initial capacity by 50 cycle eye, and, also after that, degradation progressed.

[0025] The value of x in active material $\text{LiNixMn}(1-x)\text{O}_2$ compounded using the nickel-Mn compound hydroxide adjusted according to coprecipitation has the desirable thing of the range of 0.7-0.95 so that clearly also from the above result.

[0026] Next, the value of x of positive active material expressed with general formula $\text{LiNixMn}(1-x)\text{O}_2$ performed examination which changes burning temperature about the thing of the range of 0.70-0.95. 550-degree-C process of 20 hours which is the 1st step of baking was performed like the above, and made burning temperature 550 degrees C, 600 degrees C, 650 degrees C, 700 degrees C, 750 degrees C, 800 degrees C, 850 degrees C, and 900 degrees C about subsequent baking. And the same cell as the above was constituted using such positive active material, and the charge-and-discharge cycle trial of the same conditions as the above was performed. This result is shown in drawing 3. In addition, x values in the above-mentioned formula were set to 0.8.

[0027] The initial capacity and cycle property also had the good active material which compounded burning temperature as 600 degrees C - 800 degrees C so that clearly also from drawing 3, the 550-degree C thing was inadequate in initial capacity and cycle nature, as for the 850 degrees C - 900 degrees C thing, initial capacity became a little and small, and cycle degradation also became large.

[0028] Since cycle degradation of what has a good initial capacity will become a little and large if cycle degradation will become a little and large if it becomes 800 degrees C, although the burning temperature of positive active material has 600 degrees C - 800 degrees C better than the above result, initial capacity also becomes a little and more smallish and it becomes 600 degrees C, 650 degrees C - 750 degrees C are desirable.

[0029] Although this example described the thing in case x values are 0.8, as a result of x values' performing examination about the burning temperature respectively same about the thing of the range of 0.70-0.95, the result which shows the same inclination as the case of $x= 0.8$ was obtained.

[0030] The positive active material which has the presentation of LiNi_{0.8}Mn_{0.2}O₂ was compounded using the conventional mixed formula synthesis method. First, the 2nd step of baking was carried out in 750 degrees C and 2 hours after having carried out weighing capacity of nickel hydroxide, a lithium hydroxide, and the manganese hydroxide so that the atomic ratio of nickel:Mn:Li might be set to 0.8:0.2:1.0, it mixed while the ball mill ground, and putting mixture into the alumina crucible and carrying out the 1st step of baking into oxygen in 550 degrees C and 20 hours. It cooled slowly to the after [baking] room temperature, and what was ground was used as positive active material. The X diffraction pattern of this active material did not become a single phase, but became that in which two or more phases exist. Then, as a result of performing examination which changes the 2nd step of burning temperature in 650 degrees C - 900 degrees C, a single phase came to be obtained by making burning temperature into 800 degrees C or more.

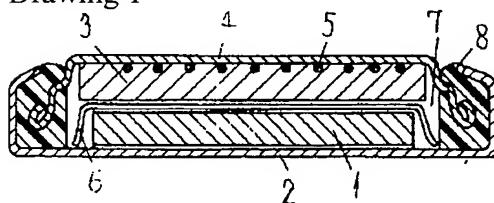
[0031] Although drawing 4 was the result of performing the same charge-and-discharge cycle trial as the above using the active material created with each above-mentioned temperature, its cycle degradation was remarkable while the crystal structure of a single phase was not acquired and the initial capacity of the positive active material which made burning temperature 650 degrees C - 750 degrees C was also small, it almost fell to 50% or less of initial capacity at the 50 cycle time, and, also after that, degradation progressed.

[0032] On the other hand, although the crystal structure became a single phase and, as for the active material compounded with the burning temperature of 800 degrees C or more, maintained 80% of initial capacity at the 50 cycle time, capacity value became 100 or less mAh/g, and decreased further.

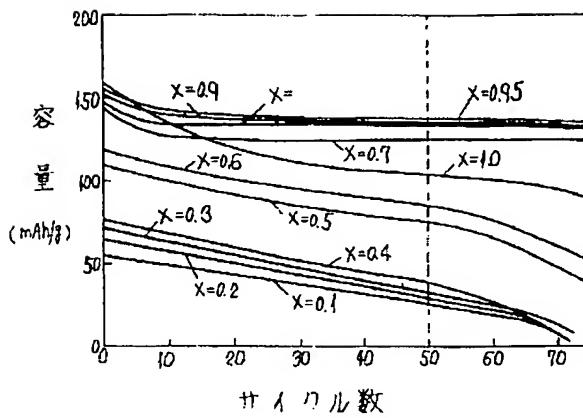
[0033]

[Effect of the Invention] As mentioned above, by the manufacturing method of the positive active material for lithium secondary batteries of this invention, since the compound hydroxide (henceforth, nickel-Mn multiple oxide) of nickel and manganese has been obtained by adding an alkali solution to the mixed solution of nickel salt and manganese salt, and carrying out coprecipitation of the hydroxide of nickel and manganese, the crystal structure has resulted in the solid-solution level which permuted some nickel certainly from Mn, almost has a single phase also by the X diffraction, and it has become what has very high crystal completeness. And if Li salt is added and calcinated to this nickel-Mn compound hydroxide, the lithium content multiple oxide which has the crystal structure which Li tends to move within a crystal can be obtained, and the positive active material capacity excelled [positive active material] in the cycle property greatly can be obtained.

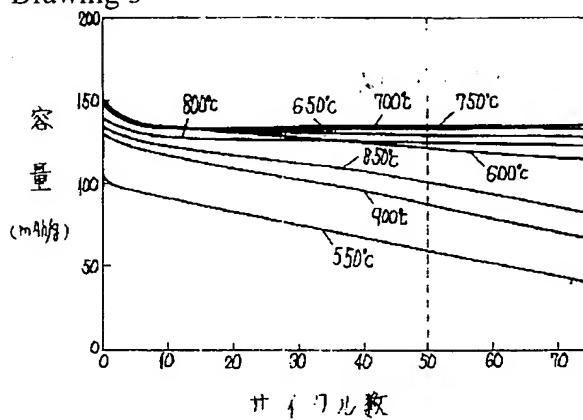
Drawing 1



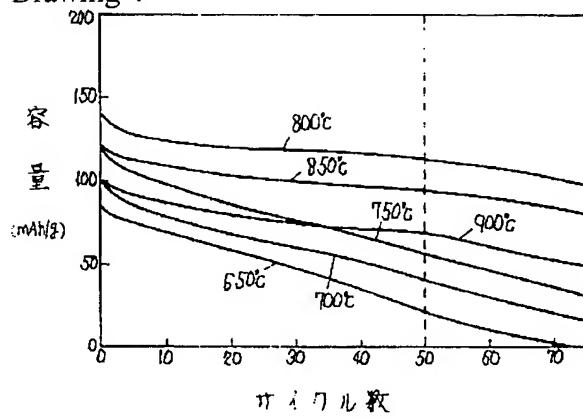
Drawing 2



Drawing 3



Drawing 4



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